ISSN: 2321-2152 IJMECE International Journal of modern

electronics and communication engineering

E-Mail editor.ijmece@gmail.com editor@ijmece.com

www.ijmece.com



InRuenceofformingconditionsonthetitaniummodelinrapidprototypingwith theselectivelasermeltingprocess

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Abstract:Inordertoevaluatethetitanium model to be used for medical purposes in rapidprototyping with these lective laser melting process, the influence of forming conditions on themechanical properties sinvestigated. The density and mechanical properties such as tensile and fatigue strengths of the model are measured. In the selective laser melting process, apulsedyttriumaluminiumgarnet (YAG) laser with average power of 50 Wandmaximumpeakpowerof3 kWisused. Thespecimens for measuring density and mechanical properties are made from commercialpuretitaniumpowders(grade1)inacontrolled atmosphere with argon gas. It is found that therelative density of the model is higher than 92 per cent and some powder particles remain within thesolidified model. The scan speed affects the tensile strength strongly and the tensile strength is around 120 percent of the standard value of the solid pure titanium when the scan speed is appropriate.However, the fatigue strengthis low, about 10 per cent of the solid one, which is still to be improved by postprocessing.

Keywords: rapid prototyping,SLM,titanium,medicalimplant,mechanicalproperty

INTRODUCTION

Rapid prototyping (RP) technology has been widely used to enhance the product development process [1]. Although this technology involves many different processes, the basic idea is a layer-by-layer additive manner in which complex-geometry models can be fabricated directly from three-dimensional computer aided design (CAD) data [2]. Geometrical and functional models, sand-casting moulds and patterns for investment casting are produced by RP. However, applications of RP to mass production of final-quality parts, so-called rapid manufacturing (RM), are still limited due to the size, surface finish and mechanical properties of the model. RM seems to be suitable for single or small lot production because of the flexibility.

One of the most promising applications for RM is in the medical area [3–5]. Dental parts, implants and

prostheses used in orthopaedic surgery are often

madeof titaniumand its alloys, because they have very goodbiocompatibility, high specific strength and an elastic mo dulus analogous to bone compared with other materials such

as Cr–Co alloys and stainless steel[6, 7].Itisalsoknownthat titaniumisdifficulttoprocessin machining and forming. In the manufactureoftitaniumimplants, e.g. artificial bones, the stan dardmodels are producedby casting, and thus the varieties of thesize and geometry are limited. Theselective lasermelting (SLM) process, in whichthe metallic powders are completely melted and clad to the already solidified base, has been proposed by the authors powdersofasingle [8–10].Inthis process, metallic compositionaresuccessively meltedinamicro-scopic zone by laser energy, which is different from theselective laser sintering

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which (SLS) method in blendedorcoatedpowdersareessential. Whenthe laserisscannedlinearly on the bed, the powder behaviour ofthe powders in **SLMis** quite differentdepending on thematerial, the shape and size of powder particles and soon. The variability of a specific material to he processedbySLMseemstodependontheflowability,thermal conductivityandsurfacetension.Forexample,

aluminiumandstainlesssteelpowdersexhibitballingphenome nain the process [**10**]. Presently, only the trialanderrormethodis usedtofind suitable materials forSLM.

Inthispaper,titaniummodels are formed with the SLM method to fabricate biomedical parts. The density and the mechanical properties such as tensile and fatiguestrengths of the titanium model are evaluated.

EXPERIMENTAL CONDITIONS

usedforthe F igure shows the SLMsystem experiment.Initiallyapowderbedwithathicknessofaround 0.1 mmisdepositedontothestainlesssteelbaseplateattached to The powderbed is scanned the piston. bv apulsedneodymium-dopedyttriumaluminiumgarnet(Nd-YAG)laserhead (LUXSTAR), which isattachedtoanx ytable. The average power of 50

Wandthemaximumpeakpowerof3

kWaresufficientenergytomelt metallic powder, andthe allows pulsedlaser manycombinationsofinteractiontimeandpeak power. Thelaserbeam can be carried through the optical fibre, and the focused laser beam diameter is 0.8 mmon thepowderbed. The first solid layer is made by the movement of thebeamon the powder bed with to arapidmeltingandsolidificationprocess inthe chambercontinuously filled with argon gas. Then the platformisloweredby0.1

mm,thenextpowderlayerisdeposited and another solid layer is made. By successive scanning and lowering of the platform a three-dimensional modelisfabricated.

The hatching pattern is shown in Fig. 2. One cycle of the hatching process for reducing distortion of the model and time is as follows: scanning only outline, scanning

outlineandhatching inside inthe xdirection,scanning onlyoutlineandscanningandhatchinginsideintheydirection.Thehatchingspaceis0.75

mmandthelayerthicknessis0.1mm.

Commercial pure titanium powder grade 1 was usedintheexperiment.Thechemicalcompositionsofthe

pure titaniumare shown in Table 1. The powder has averylowamountofhydrogen(threetimeslessthemaximu mofgrade1titaniumpowder)toavoidtheembrittlementeffe ct.Thepowderismadebytheinduction melting gas atomizing process, which leadstospherical particles. The particle diameter distributionisunder45 mmandtheaverageparticlesizeis25

mm.Theapparentdensityofthe powder is around 64 percentoftherealdensity.

SINGLE SCANNING TEST

Tofindtheoptimumformingcondition of the puretitanium power in the SLM process, the single scanningtest was carried out [**10**]. In this test, the laser beam withameanpowerof50Wwasscannedlinearlyonlyonce

topsurfaceisthemostdesirabletoformathree-

dimensionalshapeandtoimprovetheconnectionbetweenthes olidifiedlayers.F romtheaboveexperiment,thelaserirradiationconditionis determined: thepeakpower,pulsedurationandrepetitionrateare1kW,1msan d50Hzrespectively.

Usingthe determined condition, the model of bonehas been successfully made from pure titanium powdersinacontrolledatmosphere, as showninF ig. 5. InSLM, the dimension of the model is dependent on the amountofmoltenpowder and distortion of the solidified partand sintered powder sticking in the model. Although themodeldimensionisnotcontrolled strictly in the laserscan process, the titaniummodel has almost an identical dimension to the original one. The surface roughnes sof

DENSITYAND MECHANICAL PROPERTY

Specimen

Thespecimensformeasuringdensityandmechanicalproperti esofthetitanium model are shown in Fig. 6.Thecubicsampleswereused for density measurementandwereexaminedbyopticalandscanningelect ronmicroscopy(SEM).DensitywasmeasuredusingtheArchi medes principle. Evaluation of the tensile strengthwas carried out on a universal tensile testing machine ataspeedof0.5 mm/s.Torsionalfatiguetestswerecarriedout at up to 10^7 cycles in order to investigate the fatiguelimit.Theprocess

parametersofthelaserscanningareshown in Table 2 [**11**]. The influence of the peak powerandthescanspeedonthedensityandmechanicalpropert iesareexaminedintheexperiment.

Tensilestrength

The relationship betweenthe scanspeedandthe tensilestrength of the model formed with the peak kWisshowninF power of1 ig.9.Thetensilestrengthishighlydependentonthe scan speed, and there is an optimumscanspeedforthe tensile strength. The results at thescanspeed of 6 and 8 mm/s are very good(290 MPa) fortitanium parts because the tensile strengthof solid puretitaniumgrade 1is around240 MPa. Theselective lasermeltingprocessisassociatedwithsharptemperaturegra dientsthatcandecreasegrainsizeandimprovemechanical properties. The large pore size andthe high volume fraction of porosity cause low ductility of the samples.Becausetheelongationislow,flaw size andcrackpropagationareimportant[12].

Fatiguestrength

The measured fatigue strengths of the samples with thescanspeed of 6 mm/sareshowninF ig.10. The fatiguestrengthofthe titaniummodel is very low: around 10per cent of the tensile strength for 10^7 cycles. The fatiguestrengthlimitis moreaffected by porosity than tensile strength. Although tensile and fatigue strengths are notthemainprerequisitesforpuretitaniumimplants,a

fatigue strength of around 30–40 per cent of the tensilestrengthmaybenecessary, even in the seimplants. Figure 11 shows the cross-

sectionofthecubictitaniummodel.The pores have irregularshape andsharpcorners, and some powder particles remain within the solidified part without melting. It is considered that the shape of porosity and the remaining powder particles affect the fatigue strength. The effect of postprocessing, such as vacuum sintering and hot is ostatic pressing on the fatigue strength, is studied at the next stage.

DISCUSSION

Thetensile strengthof the titanium model decreases with decreasing of the scan speed from the optimum value, although the amount of melted powders increases due to a longer interaction time. The relation between the scan speed and the connection

between

solidifiedlayerscanexplainthisphenomena.Inordertoinves tigatethe connection between the solidified layersinthelasermeltingprocess,titaniumpowdersaredepo sitedonaplateofpuretitaniumgrade1thatis

modelled as a previous solidified part, and the connecting areabetween the solidified powders and the plate ismeasured.Thelayerthicknessofthepowdersandtheformin g conditions are the same as those of the previousexperiment.

Figure12showsthecross-section of the connectingareabetweenthesolidifiedpowdersandtheplate. When

thescanspeedis2

mm/s,theoverhangingshapesofthesolidifiedpartcanbeseen, andtheconnecting areabecomessmall. The connecting ratio is determined as he sum of the connecting width divided the hv hatchingwidth, which means the fraction of the solidified partjoinedtotheplate. The relationship between the scanspeedandtheconnectingratioofthesolidifiedpartisThe fatigue strength of the model is low: about 10 percentof the solid one. The cross-sectionof the modelshowspowderparticlesthat are notmeltedandremainwithinthe solidified part, which cause he low fatigue strength. Itis possible mav toimprove the fatigue strength when the relative density reaches 100percentbypost-processing.

ACKNOWLEDGEMENTS

showninFig. 13. The connecting ratioshows the sametendency as the results of the tensile strength, so that thelow tensile strength at the slow speed is due to the smallconnectingratiobetweenthesolidifiedlayers.

Also, the slow scanspeed leads to a long interactiontime on the powderbed and vaporization of the moltenpartcanoccur[13], which affects the shape of

thesolidifiedpart. By using a high-speed camera, gas fromthemoltenpartafterlaserirradiationwas

observed during the formation of a cubic model. It is also known that a plasma plume appears in all laser-

inducedprocesses involving metal vaporization. F ormationofthelow-temperatureplasmacanalsoreducethe

powerdensity on he material surface by scattering of the laserradiation.Vaporizationof

themoltenpartoraplasmaplumemay have caused the low tensile strength at theslowscanspeed.

CONCLUSIONS

Titanium models for medical purposes were formed withtheselectivelasermeltingmethodandthemechanicalpro pertiesofthemodel, such as tensile and fatiguestrength, were evaluated. The density of the model was also measured using the Archimedes principle. It is found that:

1. PuretitaniumpowderissuitablefortheSLMmethod.Itcan beformedusingthreedimensionalmodelsunderanappropriateformingcondit

the

ion.

- 2. The relative density of the model is more than 92 percent. The SLM method produces high-density titanium models.
- 3. Thetensilestrengthofthetitaniummodel ishighlydependentonthescanspeedandthereisanapprop riate speedforthe maximumtensile strength.Whenthescanspeedwas6 mm/s,thetensilestrengthofthemodelshowedthemaximumv alue(290 MPa),similartothatofsolidpuretitanium.

The authors express their thanks to Professor K. OguraandDrK.Kida,GraduateSchoolofEngineeringScienc e,OsakaUniversity,fortheir help in the fatiguetest.

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